Structure dependence of photochromism and thermochromism of azobenzene-functionalized polythiophenes

X. Y. Zhao*, M. Z. Wang

College of Material Science and Engineering, Hebei University of Science and Technology, 050054, Xinhua Road 505, Shijiazhuang, P. R. China

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Abstract. Two novel azobenzene-functionalized polythiophenes, poly[4-((4-(phenyl)azo)phenoxy)butyl-3-thienylacetate] (P4) and the copolymer of 3-hexylthiophene and 4-((4-(phenyl)azo)phenoxy)butyl-3-thienylacetate (COP64) have been synthesized. The structure dependence of photoluminescence features and thermochromic behaviors of both azobenzene-functionalized polymers was investigated. The results show that polymer structure has a strong influence on the conformation and optical properties of the resulting polythiophene derivatives. The photochemical control of photoluminescence property was achieved with homopolymer P4 using photoactive azobenzene side chains.

Keywords: smart polymers, polythiophene, photochromism, thermochromism

1. Introduction

The synthesis of 3 and 4-substituted polythiophenes has attracted much interest in the past decade because polythiophenes with suitable substituents not only will have better processibility and stability, but also may possess novel electrical, electrochemical and optical properties [1, 2]. Azobenzene chromophore was demonstrated to be good photo-isomerizable units [3–6] for reversible optical data storage and other electrooptic devices. Thus, the combination of polythiophene backbone with photoactive azobenzene groups could provide a new approach to develop other novel materials with unique electronic and optical properties. The polymers, for example, can exhibit a novel dual photochromism when the photoactive substituent was attached to the conjugated polythiophene backbone [7, 8]. Similarly functionalization of the backbone of polythiophenes with azobenzene chromophore can significantly alter electrochemical and spectroelectrochemical behavior [9] as well as conductivity [10]. Therefore, the great potential for synthesis of azobenzene-functionalized polythiophene derivatives not rely solely on their ability to compete with metals or inorganic materials in existing devices but also on their use as field-responsive materials for completely new application.

In this paper, two azobenzene-functionalized polythiophene derivatives, poly[4-((4-(phenyl)azo)phenoxy)butyl-3-thienylacetate] (P4) and the copolymer of 3-hexylthiophene and 4-((4-(phenyl)azo)phenoxy)butyl-3-thienylacetate (COP64) were synthesized (see Figure 1), the synthesis and characterization of both azobenzene-functionalized polythiophenes were also discussed. In particular, we focus our attention on the study of structure-dependent of thermochromism and photochromism, which originate from a combination of
conjugated polythiophene backbone and photoactive azobenzene group.

2. Experimental

2.1. Synthesis of 4-(4-hydroxybutyloxy)azobenzene (HBAB)

HBAB was synthesized using a similar procedure described by Reference [11]. This reaction gave a yellow power with yield 55%.

2.2. Synthesis of 3-(4-((4-phenylazo)phenoxyl)butyl)thienylacetate (ATh-4)

To a two-necked flask equipped with reflux condenser, 0.97 g (6.8 mmol) of 3-thiopheneacetic acid (Aldrich) was added under N₂ atmosphere, then 5 ml dry THF and 10 ml thionyl chloride (Aldrich) were injected into the flask with a syringe. After the solution was refluxed for 2 h, the excess THF and unreacted SOCl₂ were removed under reduced pressure at the room temperature. The solid residue was dissolved in 10 ml dry THF, and cooled down to 0–5°C by ice bath, then 1.57 g (5.8 mmol) of 4-(4-hydroxybutyloxy)azobenzene and 1 ml pyridine in 15 ml of dry THF was added. After the reaction was stirred for 5 h and the solvent was evaporated out. The product was purified by the silica gel column chromatography using chloroform as the eluant. This reaction gave a yellow crystal (yield 74%) with a melting point of 87.9°C.

FT-IR (KBr, cm⁻¹): 3116 (Ar; =CH), 2864, 2940, 2963 (aliphatic; –CH₂–), 1736 (C=O), 1603, 1581, 1474 (Ar; C=C), 1502, 1441 (N=N; cis and trans), 1248 (C–O–C; νas), 1062 (C–O–C; νa), 830 (Ar; =CH, p-substituted benzene ring), 687, 770 (Ar; =CH, mono-substituted benzene ring). ¹H NMR (CDCl₃, 400 MHz, ppm): 1.89 (m, 4H), 3.69 (s, 2H), 4.08 (t, 2H), 4.23 (t, 2H), 7.01 (d, 2H), 7.08 (s, 1H), 7.3 (d, 2H), 7.46 (s, 1H), 7.51 (t, 2H), 7.92 (m, 4H). Anal. Calcd. for C₂₂H₂₂N₂SO₃: C 67.01, H 5.58, N 7.11, S 8.12, O 12.18; Found C 67.03, H 5.56, N 7.10, S, 8.08.

2.3. Synthesis of poly[3-(4-((4-phenylazo)phenoxyl)butyl)thienylacetate] (P4)

ATh-4 was chemically oxidized with anhydrous FeCl₃ in chloroform for 5 h. It was then washed repeatedly with water and methanol, and dedoped in aqueous ammonia. Finally the polymer was further purified in a Soxhlet apparatus with methanol for 24 h.

2.4. Synthesis of copolymer of 3-(4-((4-phenylazo)phenoxyl)butyl)thienylacetate and 3-hexylthiophene (COP64)

The copolymerization was carried out by seed-polymerization method using anhydrous FeCl₃ as oxidant. In a 50 ml Schlenk reaction flask, 0.9 g (5.5 mmol) of anhydrous FeCl₃ was quickly put into the flask, which was subsequently evacuated and flushed with argon three times. Than 15 ml of anhydrous CHCl₃ was syringed into the reactor at 0–5°C followed by 0.2 g (0.51 mmol) monomer in 3 ml CHCl₃ was added in one portion. After 10 min, a solution of 3-hexylthiophene (3HT) in 5 ml of CHCl₃ was added in dropwise manner for 20 min. Then the reaction mixture was stirred at the room temperature for another period of time. The polymerization reaction was terminated by the addition of 200 ml of methanol. The precipitated polymer was then washed repeatedly with water and methanol, and dedoped in aqueous ammonia. Finally the copolymer was further purified in a Soxhlet apparatus with methanol for 24 hours.

2.5. Characterization

FT-IR spectra were recorded on a Perkin-Elmer System 2000 FT-IR spectrometer. NMR spectra were recorded on a BRUKER 400 NMR spectrometer at 400 MHz. Differential scanning calorimetry (DSC) analyses were performed at 5°C/min on a TA instruments using 2920 Modulated DSC (model 2920). Molecular weights of samples were measured by gel permeation chromatography.
GPC, Waters 410; eluent, THF) calibrated with monodispersed polystyrene standards. X-ray diffractions of polymer films on glass slides were measured using a Rigaku Rint 2000 X-ray Diffractometer. UV-visible spectra were obtained using an UV-Vis recording spectrophotometer (Shimadzu, Model UV-2501PC). For the solid-state measurements, the spectrophotometer was provided with a temperature control unit that allows measurements over a temperature range 25–250°C with a maximum error of ±2°C. Fluorescence measurements were conducted on a Shimadzu RF5000 spectrofluorophotometer. The photochromic experiments were carried out using an ARC lamp (Oriel, Model 68910, 200–500 W) as the light source. The monochromatic light peaked at 360 and 410 nm was isolated using filter (Model 58650 and 59285). To prepare thin films, the polymer was dissolved in chloroform, filtered, and then the solution was deposited by casting method onto clean quartz slides and finally dried under vacuum for 24 h.

3. Results and discussion

All polymerization processes including homopolymerization and copolymerization in our research were carried out under lower temperature and controlled addition of monomer due to the fact that α–α coupling can be considered predominant during the polymerization process carried out under these experimental conditions. The oxidative polymerization of monomer ATh-4 in the presence of FeCl3 was easy to carry out without any difficulties. For the copolymerization of ATh-4 and 3-hexylthiophene, however, in order to obtain the copolymer, the seed-oxidation polymerization technique was used. That is, the monomer ATh-4 was first initiated by FeCl3 and polymerized to form the polymer active species with moderate molecular weights, then the polymer active species was used to initiate the polymerization of 3-hexylthiophene to form copolymer. It should be also stressed that the amount of oxidant FeCl3 used in copolymerization was strictly controlled in case the formation of P3HT homopolymer in the second step.

Figure 2 illustrates the FT-IR spectra of polymer P4 and copolymer COP64, along with spectrum of the monomer ATh-4. For comparison, the spectrum of poly(3-hexylthiophene) (P3HT) is also presented in Figure 2. The polymerization and copolymerization processes are confirmed by the disappearance of the band at 745 cm⁻¹, which is attributable to C–H out-of-plane deformation vibration of thiophene rings. Spectral features of the ester derivatives of ATh-4 are clearly seen in the copolymer spectrum, especially the band at 1731 cm⁻¹ and 1252 cm⁻¹, which can be ascribable to carbonyl (C=O) stretching vibration and C–O–C stretching vibration (νas) respectively. Furthermore, the absorption bands occurring at 1378 cm⁻¹ (–CH3, δs deformation vibration) and 2956 cm⁻¹ (–CH3, νas stretching vibration) in COP64 are characteristic of 3-hexylthiophene blocks, thus the present FT-IR data support the existence of Ath-4 and 3-hexylthiophene segments in copolymer. It is necessary to mention that some of FT-IR bands associated with the polythiophene backbone strongly superimpose with those originating from the chromophore. For example, the peak at 830 cm⁻¹ (Ar; =CH, p-substituted benzene ring) in azobenzene is very close to the peak at 827 cm⁻¹ associated with the out-of-plane C–Hβ deformation vibration of thiophene rings, which is characteristic of 2,3,5-trisubstituted thiophene ring. The mole ratio of 3HT unit to ATh-4 unit in COP64 determined on the basis of the NMR result is 8.62:1 (mol/mol), which is higher than that of initial feed (5.80:1). Table 1 gives physical properties of P4 and COP64.

The effects of trans-cis isomerization of photoactive chromophore on photoluminescence properties of both polymers were investigated and the corresponding results are shown in Figure 3. Upon irradiation of the UV light of 360 nm, the photoluminescent intensity of P4 film exhibited an obvious decrease as the UV irradiation time is raised. The real origin of this result is unclear. This is probably
due to the enhancement of nonradiative decay of excited species [12, 13] caused by the conformational transition of the conjugated polythiophene backbones with the increase of irradiation time. On the other hand, the above process is also reversible, when the irradiated samples were irradiated by the visible light of 410 nm, the PL intensity of P4 film steadily increased up to the starting value before irradiation (see Figure 4). These results indicate that the optical properties of the functionalized polythiophene can be controlled reversibly by the photochemical isomerization of the azobenzene chromophores covalently attached to the thiophene rings. For copolymer COP64 film, however, the photochemical isomerization of photoactive groups in the side chains showed only a quite limited effect on photoluminescent property. (see Figure 5). When the sample COP64 film was irradiated at 360 nm by different UV irradiation time, only small variations in PL intensity were observed. Two explanations are possible; the higher crystallinity (due to the introduction of highly flexible hexyl groups) of COP64 as compared with P4 hinders the photochemical isomerization process, thus, lead to a little effect on PL property. Another explanation would be attributed to the lower content of photoactive azobenzene groups in COP64 in comparison with homopolymer P4. In other words, the trans-cis isomerization of azobenzene chromophore probably only affects the photoluminescent behavior of polythiophene conjugated units containing azobenzene side chains.

Thermochromic phenomena of neutral polythiophene derivatives may be extremely useful due to the strong blue shift in the visible region with the thermally induced disordering of the side chains [1]. The thermochromic behaviors of both polymers obtained were also investigated by heating UV-vis absorption measurements. Interestingly, no thermochromism was observed for homopolymer P4 film in addition to small variations in band absorption intensity caused by thermal cis-trans

| Table 1. Physical properties of P4 and COP64 |
| --- | --- | --- | --- | --- | --- | --- | --- |
| Samples | Morphology | $M_n$ $10^{-3}$ | $M_w$ $10^{-4}$ | $M_w/M_n$ | $T_g$ $°C$ | $T_m$ $°C$ | $T_{3\%}$b $°C$ |
| P4 | amorphous | 5.39 | 1.43 | 3.65 | 48.2 | – | 263.0 |
| COP64 | semi-crystalline | 32.5 | 5.56 | 1.68 | 22.6 | 178.5 | 241.3 |

*Measured by X-ray Diffractometer; *3% weight loss
isomerization. This can be mainly attributed to the excess number of rigid azobenzene groups present in the side chains and also the existence of more polar carbonyl groups. It is known that the thermochromic phenomenon results from a delicate balance between repulsive side chains steric interactions and attractive interchain interactions, and the latter interactions being necessary to stabilize the low-temperature planar conformation. Since their too strong side-chain steric interactions between P4 molecules, no planar (or nearly planar) conformation is accessible for this polymer at the room temperature. In other words, at low temperature P4 will be forced to adopt a nonplanar conformation due to their strong steric hindrance. Therefore, no thermally induced planar-to-nonplanar conformational transition can take place. For COP64, however, due to the introduction of thiophene segments with highly flexible and less polar hexyl side chains, a quite obvious reversible thermochromic effect has been observed in the solid (Figure 6). Upon heating, an absorption band at 560 nm related to a planar (or nearly planar) conjugated backbone showed a decrease, meanwhile a new absorption band located at 446 nm appeared and gradually increased with increasing the heating temperature. The latter is characteristic of a delocalized twisted (less conjugated) form of the polythiophene backbone [14]. This absorbance change is fully reversible, and, upon cooling, the COP64 film recovers its initial absorption spectrum. A clear isosbestic point is observed, revealing a cooperative twisting of several repeat units upon heating and the coexistence of two conformational structures (Planar and nonplanar). The thermochromic transition behavior of COP64 is quite similar to that of pure poly(3-hexylthiophene) (P3HT) (see Figure 7). This strong dependence of the thermochromic properties upon the type of substituents can be explained by a cooperative formation of intramolecular and intermolecular assemblies, which would be reversibly broken through thermally induced side chain disordering. From above results, it is clear that although the optical transitions are obviously related to conformational changes of the conjugated backbone, it seems that the polymer structure such as nature of the substituents has a strong influence on the conformation and the optical properties of the resulting polythiophene derivatives.

4. Conclusions

In this paper, the differences in thermochromism and photochromism between two azobenzene-functionalized polythiophene derivatives have been discussed and the effect of the molecular structures of azobenzene-functionalized polymers on both chromisms was also investigated. Homopolymer P4 showed a reversible photochromic behavior induced by photochemical isomerization of side chains and no thermochromism could be observed due to strong side-chain steric interactions. As thiophene segments with highly flexible and less polar hexyl side chains were introduced into polymer chains, a quite obvious reversible thermochromic effect has been observed for COP64 in the solid. However, no significant photochromism was obtained probably due to the higher crystallinity of COP64 and also the lower content of photoactive azobenzene groups in COP64. These indicated the
strong dependence of both chromisms upon the polymer structures.

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